

# Improvement of ultrathin Hafnium oxynitrides deposited on Si-substrate by adding gas ratio of $N_2/O_2$

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## *Abstract*

An investigation of the electrical and physical properties of the  $HfO_xN_y$  films in the mixtures of different gas are presented for applications in MOS devices. The  $HfO_xN_y$  films were deposited onto n-type Si substrates by sputtering using mixture of  $N_2+O_2$  and  $N_2O+O_2$ , respectively. Electrical measurements included current-voltage and capacitance-voltage measurements, and physical measurements included X-ray photoelectron spectroscopy (XPS). Films made by deposited in  $N_2+O_2$  ambient showed better leakage current behavior, demonstrating that more nitrogen content can suppress the leakage current.

*Key words:* Hafnium oxide;  $HfO_xN_y$ ; C-V; I-V; XPS

## **1. Introduction**

Because of high direct tunneling current, the use of  $SiO_2$  thinner than 2.0 nm may be quite challenging in future CMOS (complementary metal oxide semiconductor) devices [1]. As an alternative to oxide/nitride systems, much work has been done on high-K metal oxides as a means to provide a substantially thicker (physical thickness) dielectric for reduced leakage current and improved gate capacitance [2]. However most of the high-K materials e.g.  $TiO_2$  and  $Ta_2O_5$ , are not thermally stable in direct contact with Si and then

silicon oxide layers are formed at interfaces [3]. Among these dielectrics,  $HfO_xN_y$  appears to be promising due to its high-K value, thermally stable when contact with silicon substrate.  $HfO_xN_y$  shows significant reduction in leakage current density, boron penetration, superior thermal and electrical stability [4]. In this work, high quality of hafnium oxynitride ( $HfO_xN_y$ ) gate dielectric prepared by annealing at temperature  $450^\circ C$  and sputtering in different gas mixture of  $N_2O + O_2$  and  $N_2 + O_2$  are compared.

## 2. Experimental

MIS capacitor structures with  $\text{HfO}_x\text{N}_y$  dielectrics were fabricated on silicon substrates. The n type (100) silicon wafers were cleaned with phosphorous acid, ethanol and deionized wafer, and then etched with HF solution ( $\text{CH}_3\text{CH}_2\text{OH}:\text{HF} = 4:1$ ).  $\text{HfO}_x\text{N}_y$  thin films were deposited through two steps [5]. First, Hf metal was pre-deposited in  $\text{N}_2\text{O} + \text{O}_2$  ambient and then a thin  $\text{HfO}_x\text{N}_y$  layer was deposited in  $\text{N}_2 + \text{O}_2$  ambient from Hf metal target at room temperature with a power of 100 W. These samples were then annealed in  $\text{N}_2$  ambient using an annealing furnace at temperature  $450^\circ\text{C}$  for 30 minutes. Al sputtering was carried out at room temperature and a power of 100 W for 30 minutes. The samples were sputtered at three different kinds of deposition pressure, 10 mtorr, 20 mtorr, and 30 mtorr.

## 3. Result and Discussion

We examined the characteristics of  $\text{HfO}_x\text{N}_y$  films deposited by two different kinds of gas mixture in three different kinds of deposition pressure, as shown in Fig. 1(a) and (b).

The equivalent oxide thickness (EOT) of  $\text{HfO}_x\text{N}_y$  films was about  $14.5\text{\AA}$  (30 mtorr) to  $25\text{\AA}$  (10 mtorr) in  $\text{N}_2\text{O} + \text{O}_2$  ambient and  $15.9\text{\AA}$  (30 mtorr) to  $27.7\text{\AA}$  (10 mtorr) in  $\text{N}_2 + \text{O}_2$  ambient. We can find that the EOT was reduced with increasing of deposition pressure.

We also observed that the capacitance of  $\text{HfO}_x\text{N}_y$  films in  $\text{N}_2\text{O} + \text{O}_2$  ambient were similar to those in  $\text{N}_2 + \text{O}_2$  ambient. In the Fig. 2(a) and (b) show the  $I$ - $V$  curves of Al/ $\text{HfO}_x\text{N}_y$ /Si structure. There exists a significant difference in leakage current.

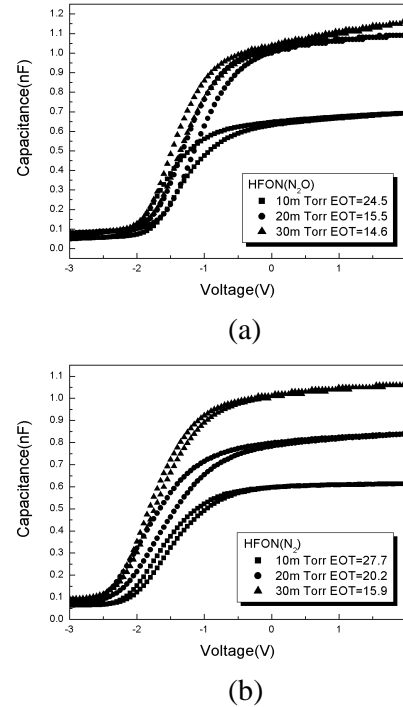
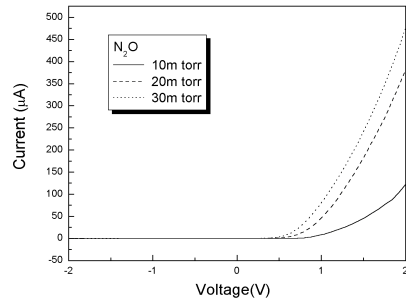
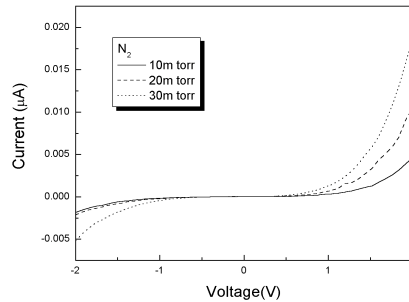


Fig. 1. The measurement of C-V curves at different mixtures of gas (a)  $\text{N}_2\text{O} + \text{O}_2$  and (b)  $\text{N}_2 + \text{O}_2$ .

The samples deposited in  $\text{N}_2 + \text{O}_2$  ambient show well-behaved leakage current characteristic. The leakage current in  $\text{N}_2 + \text{O}_2$  ambient at 1.5 V was about  $5.86 \times 10^{-9} \text{ A/cm}^2$ , which was about three orders smaller than that in  $\text{N}_2\text{O} + \text{O}_2$  ambient. Consequently, we were able to confirm that the leakage current of  $\text{HfO}_x\text{N}_y$  films deposited in  $\text{N}_2 + \text{O}_2$  ambient was improved.



(a)



(b)

Fig. 2. The measurement of I-V curves at different mixtures of gas (a)  $N_2O+O_2$  and (b)  $N_2+O_2$ .

Fig. 3. (a) and (b) show the O1s features by using the different gas ratios of  $N_2/O_2$  and  $N_2O/O_2$ , respectively. From fig.3.(a), we clearly observed the peak of silicate bonding at around 533 eV, which is associated with the bonding of Si-O-Si and Hf-O-Si rather than the metal oxide bonding. By increasing the deposition pressures, the intensity of peak position is also enhanced. On the contrary, as shown in fig. 3(b), by increasing the deposition pressures, the intensity of peak is reduced. A similar trend is also shown in fig. 4(a) and (b) for the Si 2p features, which exhibit a feature of double peak in the spectra. The position of the major peak is close to the Si-O bonding at 103.3 eV, but

there is no adequate signal for the Si-N bonding to be detected. In fact, a small signal was still detected in N1s spectra (which is not shown in this paper), in which nitrogen play important role in improving the electrical characteristics. It is suggested that the nitrogen signal need to be further studied on the annealed  $HfO_xN_y$  samples. As shown in fig. 4 (b), we observe that the intensity of the peak decrease with the increasing of the deposition pressures.

#### 4. Conclusion

In this paper, we have compared the electrical characteristics of  $HfO_xN_y$  thin films at different gas ratios for mixtures of  $N_2O+O_2$  and  $N_2+O_2$ . The leakage current of films in  $N_2+O_2$  ambient was three order less than those in  $N_2O+O_2$  ambient. However, the leakage current increases with the increasing of deposition pressures. Nonetheless, the XPS spectra of Si 2p peaks have no adequate signal to confirm the existed of Si-N bonding. Further investigation on the  $HfO_xN_y$  samples should be performed in order to confirm the incorporation of nitrogen in the  $HfO_xN_y$  films.

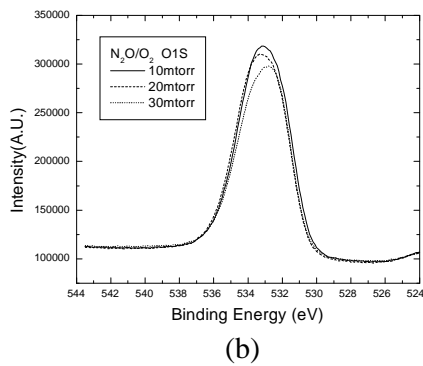
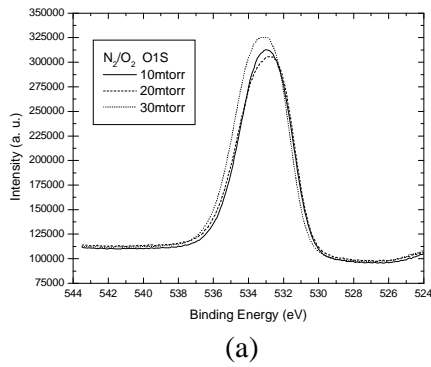


Fig. 3. XPS spectra of O1s for  $\text{HfO}_x\text{N}_y$  films with the (a)  $\text{N}_2$  and  $\text{O}_2$  (b)  $\text{N}_2\text{O}$  and  $\text{O}_2$  forming gases was performed at room temperature

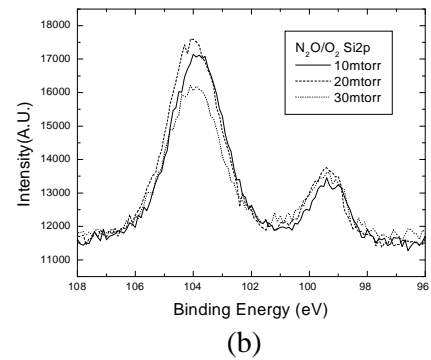
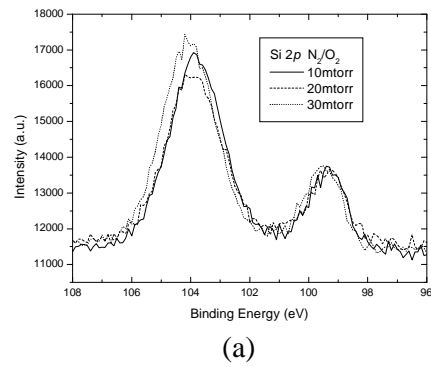


Fig. 4. XPS spectra of Si2p for  $\text{HfO}_x\text{N}_y$  films with the (a)  $\text{N}_2$  and  $\text{O}_2$  (b)  $\text{N}_2\text{O}$  and  $\text{O}_2$  forming gases was performed at room temperature

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